



# “Mixed” Tm:Ca(Gd,Lu)AlO<sub>4</sub> — a novel crystal for tunable and mode-locked 2 μm lasers

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**Abstract:** We report on the crystal growth, spectroscopy characterization and first laser operation of a new tetragonal disordered “mixed” calcium aluminate crystal, Tm:Ca(Gd,Lu)AlO<sub>4</sub>. The introduction of Lu<sup>3+</sup> leads to an additional inhomogeneous broadening of Tm<sup>3+</sup> absorption and emission spectra compared to the well-known Tm:CaGdAlO<sub>4</sub>. The maximum stimulated-emission cross-section for the <sup>3</sup>F<sub>4</sub> → <sup>3</sup>H<sub>6</sub> Tm<sup>3+</sup> transition is  $0.91 \times 10^{-20}$  cm<sup>2</sup> at 1813 nm for σ-polarization, and the emission bandwidth is more than 200 nm. A continuous-wave diode-pumped Tm:Ca(Gd,Lu)AlO<sub>4</sub> laser generates 1.82 W at 1945 nm with a slope efficiency of 29%. Under Ti:Sapphire laser pumping, a continuous tuning of the laser wavelength from 1836 to 2083 nm (tuning range: 247 nm) is demonstrated. The Tm:Ca(Gd,Lu)AlO<sub>4</sub> crystal is promising for tunable/femtosecond lasers at ~2 μm.

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## 1. Introduction

Tetragonal (space group *I4/mmm*) calcium rare-earth aluminates CaLnAlO<sub>4</sub> (where Ln = Gd or Y, abbreviated as CALGO and CALYO, respectively) are well-known disordered laser crystal hosts for Yb<sup>3+</sup>-doping [1,2]. They feature high thermal conductivity (~6.5 W/mK for CaGdAlO<sub>4</sub>) with a weak concentration-dependence and weak and positive thermal lensing originating from negative thermo-optic coefficients,  $dn/dT$  [3]. The structural disorder originates from a random distribution of Ca<sup>2+</sup> and Ln<sup>3+</sup> cations over a single type of sites (C<sub>4v</sub>) [2]. This leads to a significant inhomogeneous broadening of absorption and emission bands making these crystals attractive for sub-100 fs mode-locked (ML) lasers at ~1 μm in the case of Yb<sup>3+</sup> doping [4,5]. Good thermo-mechanical properties of CaLnAlO<sub>4</sub>-type crystals enable power scaling of Yb<sup>3+</sup> oscillators [6].

Considering the success of Yb:CaLnAlO<sub>4</sub> crystals, the research interest turned to their doping with Thulium (Tm<sup>3+</sup>) ions. The latter are known for their eye-safe emission in the spectral range of ~2 μm (<sup>3</sup>F<sub>4</sub> → <sup>3</sup>H<sub>6</sub> transition) [7]. The Tm<sup>3+</sup> ions show efficient absorption at ~0.8 μm (to the <sup>3</sup>H<sub>4</sub> state), e.g., emission from Ti:Sapphire lasers or commercial powerful AlGaAs laser diodes. Efficient cross-relaxation for adjacent Tm<sup>3+</sup> ions, <sup>3</sup>H<sub>4</sub> + <sup>3</sup>H<sub>6</sub> → <sup>3</sup>F<sub>4</sub> + <sup>3</sup>F<sub>4</sub>, may raise the pump quantum efficiency up to 2 leading to high laser efficiency and reduced heat loading [8]. Due to the typically large Stark splitting of the ground-state (<sup>3</sup>H<sub>6</sub>), the ~2 μm

Tm<sup>3+</sup> emission band is broad opening the possibility of broadband wavelength tuning of the laser emission and ML laser operation.

In 1990s, Tm:CaYAlO<sub>4</sub> crystals were initially studied [9,10]. In the recent years, Tm<sup>3+</sup>-doped CaLnAlO<sub>4</sub> crystals were revisited regarding their growth, spectroscopic properties [11–13] and continuous-wave (CW) laser emission at ~2 μm [12,14]. In [14], a diode-pumped Tm:CaYAlO<sub>4</sub> laser generated a maximum output power of 7.62 W at 1945 nm with a slope efficiency of 50.9% (vs. the absorbed pump power). Passive Q-switching of such lasers by Cr<sup>2+</sup>:ZnSe saturable absorbers has been reported [15]. Recently, an in-band pumping scheme (at ~1.7 μm, directly to the upper laser level, <sup>3</sup>F<sub>4</sub>) was explored for Tm:CaYAlO<sub>4</sub> leading to the generation of 6.8 W at 1968 nm with a slope efficiency of 52% (vs. the incident pump power) [16].

Regarding ML Tm:CaLnAlO<sub>4</sub> lasers, a remarkable result is the generation of femtosecond pulses (650 fs) from a Tm:CaGdAlO<sub>4</sub> laser ML by a GaSb-based Semiconductor Saturable Absorber Mirror (SESAM) [17]. The central wavelength of the laser spectrum was at 2021 nm with a full width at half maximum (FWHM) of 9.2 nm. Such a long laser wavelength (for Tm<sup>3+</sup>-doped crystals) was because a special “bandpass” output coupler was employed and the laser was constrained to oscillate above 2 μm. This had a key effect for achieving fs pulses. In [18], a SESAM ML Tm:CaYAlO<sub>4</sub> laser operating at 1961 nm generated pulses with a duration as long as 35.3 ps.

Note that for most of the laser materials doped with Tm<sup>3+</sup>, the center of the <sup>3</sup>F<sub>4</sub> → <sup>3</sup>H<sub>6</sub> emission band is located slightly below 2 μm. This is unfavorable for ML because the laser emission spectrally overlaps with the atmosphere water vapor absorption bands preventing fs mode-locking. One way to solve this problem is to use special wavelength-selective cavity mirrors as described above to constrain the laser to operate at >2 μm (the corresponding part of the gain spectrum is related to the electron-phonon (vibronic) interaction [19]). Another option is to use a very limited number of host materials ensuring large Stark splitting of the <sup>3</sup>H<sub>6</sub> Tm<sup>3+</sup> ground-state (e.g., cubic sesquioxides [20]). Recently, a new monoclinic Tm:MgWO<sub>4</sub> crystal generating laser emission above 2 μm enabled the generation of 86 fs pulses at 2017 nm [21]. An alternative is to provide an additional inhomogeneous broadening of Tm<sup>3+</sup> emission by structure disorder or compositional disorder (in a “mixed” crystal) of the host material affecting the crystal field. Recently, various Tm<sup>3+</sup>-doped crystals and ceramics with structure or compositional disorder were studied for fs ML lasers at ~2 μm [22,23].

In the present work, we aimed to grow and study the spectroscopic and laser properties of a novel calcium aluminate crystal, Tm:Ca(Gd,Lu)AlO<sub>4</sub>. The addition of optically passive Lu<sup>3+</sup> ions is expected to provide additional compositional disorder leading to further spectral broadening which is attractive for fs pulse generation in ML lasers. Note that the existence of the CaLuAlO<sub>4</sub> crystal has never been reported so far. The growth and spectroscopy of an Yb:Ca(Gd,Lu)AlO<sub>4</sub> crystal was recently studied [24].

## 2. Crystal growth and structure

The Tm:Ca(Gd,Lu)AlO<sub>4</sub> crystal melts congruently and thus it was grown by the conventional Czochralski (Cz) method using an Ar atmosphere in an Ir crucible. An automatic system was used to control the boule diameter. The polycrystalline samples were obtained by solid-state reaction from a mixture of the starting materials, 4N-pure CaCO<sub>3</sub>, Gd<sub>2</sub>O<sub>3</sub>, Lu<sub>2</sub>O<sub>3</sub>, Al<sub>2</sub>O<sub>3</sub> and 5N-pure Tm<sub>2</sub>O<sub>3</sub>. They were placed in an Ir crucible and melted by an intermediate-frequency heater. A [001]-oriented CaGdAlO<sub>4</sub> seed was used, the pulling rate was 0.5 mm/h and the rotation speed was 8 rpm. After the growth was completed, the crystal was cooled down to room temperature at a stepping rate of 15-25 K/h. A crack-free large-volume as-grown boule, Fig. 1(a), had a yellow coloration, attributed to interstitial oxygen atoms. The coloration was removed to a great extent by annealing at 950 °C for 24 h under N<sub>2</sub> atmosphere with 5% H<sub>2</sub>.

The concentration of doping ions was measured by Inductively Coupled Plasma Mass Spectrometry (ICP-MS) to be 3.5 at.% Tm<sup>3+</sup> and 5.2 at.% Lu<sup>3+</sup>. The segregation coefficients

$K = N_{\text{crystal}}/N_{\text{melt}}$  for these ions are 0.70 and 0.52, respectively. The stoichiometric crystal formula is  $\text{CaGd}_{0.913}\text{Lu}_{0.052}\text{Tm}_{0.035}\text{AlO}_4$ . The  $\text{Tm}^{3+}$  ion concentration is  $4.31 \times 10^{20}/\text{cm}^3$  (crystal density:  $\rho = 6.01 \text{ g/cm}^3$ ).

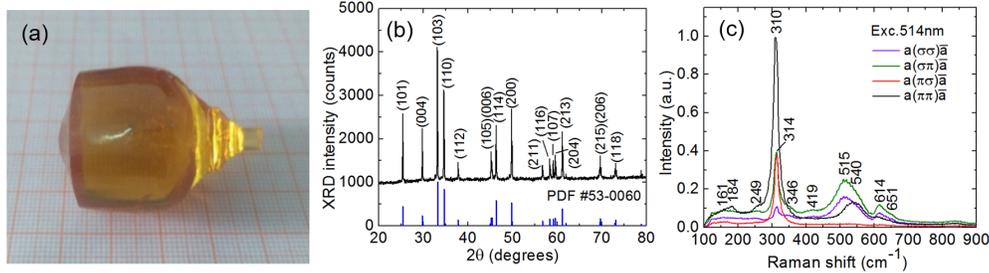


Fig. 1. (a) Photograph of the as-grown  $\text{Tm}:\text{Ca}(\text{Gd},\text{Lu})\text{AlO}_4$ ; (b) X-ray powder diffraction (XRD) pattern (in black), numbers denote the Miller's indices ( $hkl$ ), standard XRD pattern of  $\text{CaGdAlO}_4$  is shown for comparison (in blue); (c) polarized Raman spectra of an  $a$ -cut crystal,  $a(xy)a$  are the Porto's notations, numbers denote the peak Raman frequencies in  $\text{cm}^{-1}$ ,  $\lambda_{\text{exc}} = 514 \text{ nm}$ .

In  $\text{CaGdAlO}_4$ , the  $\text{Ca}^{2+}$  and  $\text{Gd}^{3+}$  cations (ionic radii  $R_{\text{Ca}} = 1.180 \text{ \AA}$ ,  $R_{\text{Gd}} = 1.107 \text{ \AA}$ ) statistically occupy the same type of site (Wyckoff symbol:  $4e$ , site symmetry:  $C_{4v}$ , coordination number: IX).  $\text{Tm}^{3+}$  ( $R_{\text{Tm}} = 1.052 \text{ \AA}$ ) and  $\text{Lu}^{3+}$  ( $R_{\text{Lu}} = 1.032 \text{ \AA}$ ) cations are entering these sites. The local disorder results from the second coordination sphere of  $\text{Tm}^{3+}$  ions constituted by  $\text{Gd}^{3+}|\text{Lu}^{3+}$  and  $\text{Ca}^{2+}$  ones, namely the charge difference of these ions and the different cation-cation distances. Note that doping of  $\text{Ca}(\text{Gd},\text{Lu})\text{AlO}_4$  by both  $\text{Tm}^{3+}$  and  $\text{Lu}^{3+}$  ions is expected to affect the crystal field because of the closeness of their ionic radii. In this way, one can argue that solely high  $\text{Tm}^{3+}$  doping can produce a similar effect to  $\text{Tm}^{3+}$ ,  $\text{Lu}^{3+}$  codoping. In the latter case, first, better crystal quality is achieved. Moreover, one avoids the unwanted energy-transfer upconversion related to very high  $\text{Tm}^{3+}$  concentration.

The structure and phase purity of  $\text{Tm}:\text{Ca}(\text{Gd},\text{Lu})\text{AlO}_4$  were determined by X-ray powder diffraction (XRD), Fig. 1(b). The crystal is tetragonal (space group  $I4/mmm - D_{4h}^{17}$ , No. 139); the lattice constants are  $a = 3.6446 \text{ \AA}$  and  $c = 12.2157 \text{ \AA}$ . The XRD pattern is in agreement with that for undoped  $\text{CaGdAlO}_4$ .

The vibronic properties of an  $a$ -cut crystal were studied by Raman spectroscopy, Fig. 1(c). The most intense Raman peak is found at  $310 \text{ cm}^{-1}$ . It is red-shifted with respect to undoped  $\text{CaGdAlO}_4$  ( $330 \text{ cm}^{-1}$ ) indicating a structure modification. The factor group analysis of the  $D_{4h}^{17}$  unit cell predicts the following irreducible representations ( $\mathbf{k} = 0$ ):  $\Gamma = 2A_{1g} + 2E_g + 4A_{2u} + 5E_u + B_{2u}$  of which the  $2A_{1g}$  and  $2E_g$  modes are Raman-active. The band at  $310 \text{ cm}^{-1}$  is assigned as  $E_g$ . The highest phonon frequency  $h\nu_{\text{ph}}$  is  $651 \text{ cm}^{-1}$ .

### 3. Optical spectroscopy

$\text{Tm}:\text{Ca}(\text{Gd},\text{Lu})\text{AlO}_4$  crystals are optically uniaxial (the optical axis is parallel to the  $c$ -axis). There are two principal light polarizations,  $\mathbf{E} \parallel c$  ( $\pi$ ) and  $\mathbf{E} \perp c$  ( $\sigma$ ). All spectroscopic studies were performed using an  $a$ -cut crystal at room-temperature (RT, 293 K).

The RT absorption spectra of  $\text{Tm}:\text{Ca}(\text{Gd},\text{Lu})\text{AlO}_4$  for  $\pi$  and  $\sigma$  polarizations are shown in Fig. 2(a). The broad absorption in the visible (300-550 nm) is related to the residual absorption of color centers [25]. The sharp band at 310 nm is due to the  $\text{Gd}^{3+}$  ions (the  ${}^8S_{7/2} \rightarrow {}^6P_{7/2}$  transition). Note that the  $\text{CaLnAlO}_4$  crystals have an indirect bandgap  $E_g$  of  $\sim 4.2 \text{ eV}$  [26] corresponding to the UV absorption edge at  $\sim 295 \text{ nm}$ . In the spectrum, the bands related to  $\text{Tm}^{3+}$  transitions from the ground-state ( ${}^3H_6$ ) to the excited ones (from  ${}^3F_4$  up to  ${}^1D_2$ ) are clearly resolved. The details about the  ${}^3H_6 \rightarrow {}^3H_4$  transition suitable for diode-pumping are shown in Fig. 2(b). The maximum absorption cross-section  $\sigma_{\text{abs}} = \alpha_{\text{abs}}/N_{\text{Tm}}$  of  $1.66 \times 10^{-20} \text{ cm}^2$  at 792.3 nm corresponds to  $\pi$ -polarization and the full width at half maximum (FWHM) of the

corresponding absorption peak is 17.5 nm. This is broader than for the 1.8 at.% Tm:CaGdAlO<sub>4</sub> crystal studied for comparison (16.3 nm). For  $\sigma$ -polarization,  $\sigma_{\text{abs}}$  is  $0.67 \times 10^{-20} \text{ cm}^2$  at 798.1 nm.

The absorption spectra were analyzed within the standard Judd-Ofelt (J-O) theory [27,28], Table 1. The squared reduced matrix elements for Tm<sup>3+</sup> ions  $U^{(k)}$  were taken from [29]. The J-O (intensity) parameters are  $\Omega_2 = 2.933$ ,  $\Omega_4 = 2.787$  and  $\Omega_6 = 1.413 [10^{-20} \text{ cm}^2]$ .

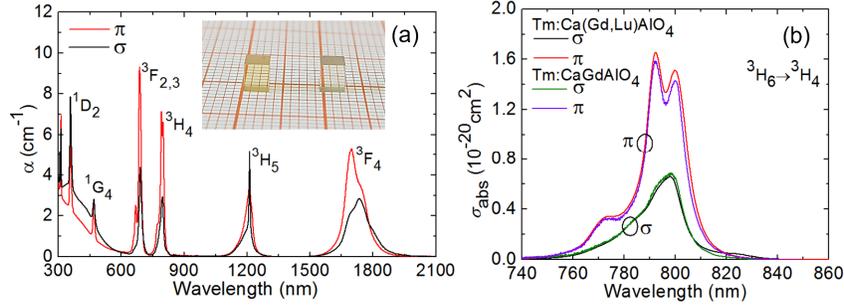


Fig. 2. Absorption of Tm:Ca(Gd,Lu)AlO<sub>4</sub>: (a) absorption spectrum of the annealed crystal ( $\alpha$ : absorption coefficient), *inset* – photograph of the sample before (*left*) and after (*right*) annealing; absorption cross-sections,  $\sigma_{\text{abs}}$ , for the  ${}^3\text{H}_6 \rightarrow {}^3\text{H}_4$  Tm<sup>3+</sup> transition. In (b), the spectra for 1.8 at.% Tm:CaGdAlO<sub>4</sub> are also shown. The light polarization is denoted by  $\pi$  and  $\sigma$ .

**Table 1. Experimental and Calculated Absorption Oscillator Strengths for Tm:Ca(Gd,Lu)AlO<sub>4</sub>**

Transition	$\langle \lambda \rangle$ , nm	$\langle E \rangle$ , cm <sup>-1</sup>	$\langle n \rangle$	$\langle I \rangle$ , cm <sup>-1</sup> nm	$\langle f \rangle_{\Sigma}^{\text{exp}} \times 10^6$	$\langle f \rangle_{\Sigma}^{\text{calc}} \times 10^6$
${}^3\text{H}_6 \rightarrow {}^3\text{F}_4$	1741.8	5741	1.912	407.86	3.523	3.506 <sup>ED</sup>
${}^3\text{H}_6 \rightarrow {}^3\text{H}_5$	1203.4	8310	1.921	131.01	2.371	2.417 <sup>ED</sup> + 0.527 <sup>MD</sup>
${}^3\text{H}_6 \rightarrow {}^3\text{H}_4$	794.3	12590	1.933	94.56	3.928	3.656 <sup>ED</sup>
${}^3\text{H}_6 \rightarrow {}^3\text{F}_2 + {}^3\text{F}_3$	687.6	14544	1.939	108.58	6.02	5.618 <sup>ED</sup>
${}^3\text{H}_6 \rightarrow {}^1\text{G}_4$	490.7	20378	1.964	14.05	1.529	1.213 <sup>ED</sup>
${}^3\text{H}_6 \rightarrow {}^1\text{D}_2$	362.6	27576	2.009	24.33	4.849	4.685 <sup>ED</sup>
rms deviation						0.69

$\langle \lambda \rangle$  - “center of gravity” of the absorption band,  $\langle E \rangle$  - energy barycenter of the multiplet,  $\langle n \rangle$  - mean refractive index,  $\langle I \rangle$  - integrated absorption coefficient,  $f_{\Sigma}^{\text{exp}}$  and  $f_{\Sigma}^{\text{calc}}$  - experimental and calculated absorption oscillator strengths (ED + MD), respectively. The  $\langle \rangle$  brackets indicate polarization-averaging,  $(2\sigma + \pi)/3$ .

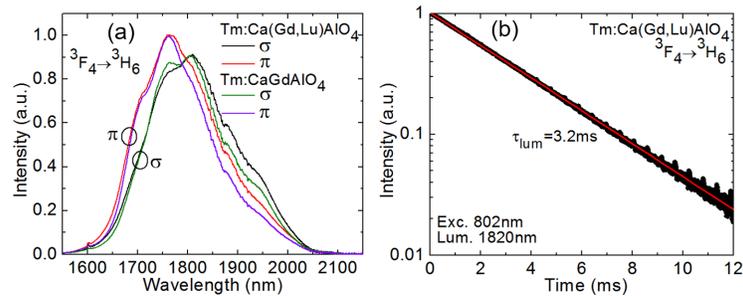


Fig. 3. Luminescence of Tm<sup>3+</sup> in Ca(Gd,Lu)AlO<sub>4</sub>: (a) luminescence spectra for the  ${}^3\text{F}_4 \rightarrow {}^3\text{H}_6$  transition,  $\lambda_{\text{exc}} = 802 \text{ nm}$ ; (b) luminescence decay curve, *circles* – experimental data, *line* – single-exponential fit. In (a), the spectra for Tm:CaGdAlO<sub>4</sub> are given for comparison. The light polarization is denoted by  $\pi$  and  $\sigma$ .

The polarized luminescence spectra of the Tm:Ca(Gd,Lu)AlO<sub>4</sub> crystal (the  ${}^3\text{F}_4 \rightarrow {}^3\text{H}_6$  Tm<sup>3+</sup> transition) are shown in Fig. 3(a). The emission spectra are smooth and broad. The highest emission intensity corresponds to  $\pi$ -polarization. The FWHM of the emission

spectrum is 187 ( $\pi$ ) and 207 ( $\sigma$ ) nm which is broader than for Tm:CaGdAlO<sub>4</sub>, 168 ( $\pi$ ) and 186 ( $\sigma$ ) nm. Thus, the introduction of Lu<sup>3+</sup> ions leads to an additional spectral broadening due to the compositional disorder.

The luminescence decay of Tm<sup>3+</sup> ions from the <sup>3</sup>F<sub>4</sub> multiplet was studied using the pinhole method [30] to diminish the effect of reabsorption on the measured lifetime. The decay curve corresponding to the smallest pinhole (diameter: ~100  $\mu$ m) is shown in Fig. 3(b). It is clearly single-exponential; the decay time  $\tau_{\text{lum}}$  is 3.2 ms.

The probabilities of spontaneous radiative transitions  $A_{\Sigma}(JJ')$ , the luminescence branching ratios  $B(JJ')$  and the radiative lifetimes  $\tau_{\text{rad}}$  were calculated using the J-O theory, Table 2. For the upper laser level (<sup>3</sup>F<sub>4</sub>),  $\tau_{\text{rad}} = 2.46$  ms. This value is shorter than the measured lifetime. The observed difference is attributed to the hypersensitivity of the <sup>3</sup>H<sub>6</sub>  $\rightarrow$  <sup>3</sup>F<sub>4</sub> transition of Tm<sup>3+</sup> ions [31] affecting the precision of the J-O calculations. For the <sup>3</sup>H<sub>4</sub> state,  $\tau_{\text{rad}} = 0.44$  ms.

Table 2. Calculated Emission Probabilities for Tm<sup>3+</sup> in Tm:Ca(Gd,Lu)AlO<sub>4</sub>

Excited state	Terminal state	$\langle\lambda\rangle$ , nm	$A_{\Sigma}^{\text{calc}}(JJ')$ , s <sup>-1</sup>	$B(JJ')$ , %	$A_{\text{tot}}$ , s <sup>-1</sup>	$\tau_{\text{rad}}$ , ms
<sup>3</sup> F <sub>4</sub>	<sup>3</sup> H <sub>6</sub>	1741.8	407.2 <sup>ED</sup>	100	407.2	2.46
<sup>3</sup> H <sub>5</sub>	<sup>3</sup> F <sub>4</sub>	3893.2	13.6 <sup>ED</sup> + 1.3 <sup>MD</sup>	2.5	600.3	1.66
	<sup>3</sup> H <sub>6</sub>	1203.4	485.6 <sup>ED</sup> + 99.8 <sup>MD</sup>	97.5		
<sup>3</sup> H <sub>4</sub>	<sup>3</sup> H <sub>5</sub>	2336.5	58.4 <sup>ED</sup> + 12.2 <sup>MD</sup>	3.0	2366.8	0.42
	<sup>3</sup> F <sub>4</sub>	1460.2	182.6 <sup>ED</sup> + 27.8 <sup>MD</sup>	8.9		
	<sup>3</sup> H <sub>6</sub>	794.3	2085.8 <sup>ED</sup>	88.1		
<sup>3</sup> F <sub>2</sub> + <sup>3</sup> F <sub>3</sub>	<sup>3</sup> H <sub>4</sub>	5118.8	7.9 <sup>ED</sup> + 0.2 <sup>MD</sup>	0.2	5229.4	0.19
	<sup>3</sup> H <sub>5</sub>	1604.2	474.2 <sup>ED</sup>	9.0		
	<sup>3</sup> F <sub>4</sub>	1136.1	161.8 <sup>ED</sup> + 71.2 <sup>MD</sup>	4.5		
	<sup>3</sup> H <sub>6</sub>	687.6	4514.1 <sup>ED</sup>	86.3		
<sup>1</sup> G <sub>4</sub>	<sup>3</sup> F <sub>2</sub> + <sup>3</sup> F <sub>3</sub>	1713.6	92.8 <sup>ED</sup> + 4.1 <sup>MD</sup>	2.6	3744.6	0.27
	<sup>3</sup> H <sub>4</sub>	1283.8	259.4 <sup>ED</sup> + 40.1 <sup>MD</sup>	8.0		
	<sup>3</sup> H <sub>5</sub>	828.5	980.7 <sup>ED</sup> + 183.6 <sup>MD</sup>	31.1		
	<sup>3</sup> F <sub>4</sub>	683.2	301.1 <sup>ED</sup> + 11.1 <sup>MD</sup>	8.3		
	<sup>3</sup> H <sub>6</sub>	490.7	1871.7 <sup>ED</sup>	50.0		

$\langle\lambda\rangle$  - mean emission wavelength estimated from the barycenter energies of multiplets, cf. Table 1,  $A_{\Sigma}^{\text{calc}}$  - probability of radiative spontaneous transitions (ED + MD),  $B(JJ')$  - luminescence branching ratio,  $A_{\text{tot}}$  and  $\tau_{\text{rad}}$  - total probability of radiative spontaneous transitions and radiative lifetime of the excited state, respectively. The <sup>3</sup>F<sub>3</sub> + <sup>3</sup>F<sub>2</sub> multiplets are considered as thermally coupled states.

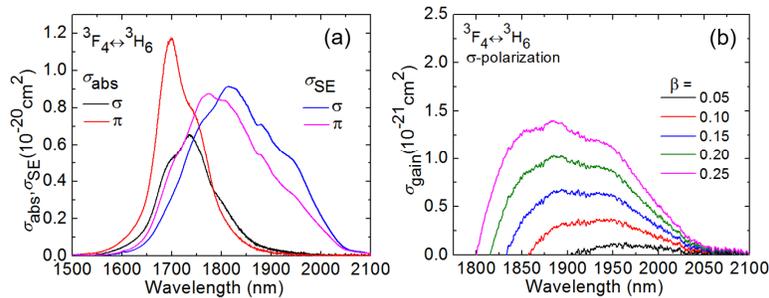


Fig. 4. Transition cross-sections of Tm<sup>3+</sup> in Ca(Gd,Lu)AlO<sub>4</sub> crystal: (a) absorption,  $\sigma_{\text{abs}}$ , and stimulated-emission,  $\sigma_{\text{SE}}$ , cross-sections for the <sup>3</sup>F<sub>4</sub>  $\leftrightarrow$  <sup>3</sup>H<sub>6</sub> transition ( $\pi$  and  $\sigma$  light polarization); (b) gain cross-sections,  $\sigma_{\text{gain}} = \beta\sigma_{\text{SE}} - (1 - \beta)\sigma_{\text{abs}}$ , for various inversion ratios  $\beta = N_2(^3F_4)/N_{\text{Tm}}$  ( $\sigma$  light polarization).

The stimulated-emission (SE) cross-sections,  $\sigma_{\text{SE}}$ , for the <sup>3</sup>F<sub>4</sub>  $\rightarrow$  <sup>3</sup>H<sub>6</sub> transition and  $\pi$  and  $\sigma$  polarizations were calculated using a combination of the modified reciprocity method (mRM) [32] and the Fuchtbauer–Ladenburg (F-L) equation [33], see Fig. 4(a). The refractive indices were taken from [34],  $n_o = 1.904$  and  $n_e = 1.927$  at ~1.8  $\mu$ m. The maximum  $\sigma_{\text{SE}} = 0.91 \times 10^{-20}$  cm<sup>2</sup> at 1813 nm for  $\sigma$ -polarization. As Tm<sup>3+</sup> ions represent a quasi-three-level laser scheme,

the laser emission is expected at longer wavelengths. For a local peak in the SE cross-section spectra at 1946 nm for  $\sigma$ -polarization,  $\sigma_{SE}$  is  $0.50 \times 10^{-20} \text{ cm}^2$ .

The gain cross-sections,  $\sigma_{\text{gain}} = \beta\sigma_{SE} - (1 - \beta)\sigma_{\text{abs}}$ , where  $\beta = N_2(^3F_4)/N_{Tm}$  is the inversion ratio are plotted in Fig. 4(b) for  $\sigma$ -polarization. For small inversion ratios  $\beta < 0.10$ , the laser emission is expected at  $\sim 1950 \text{ nm}$ . For  $\beta = 0.15$ , the gain bandwidth is 145 nm.

#### 4. Laser operation

First, we studied CW operation of Tm:Ca(Gd,Lu)AlO<sub>4</sub> crystal under diode-pumping without any wavelength-selective element in the laser cavity. The active element was *a*-cut (thickness:  $t = 3.4 \text{ mm}$ , aperture:  $3.0(c) \times 3.0 \text{ mm}^2$ ). It was polished to laser quality, wrapped with In foil and mounted in a water-cooled ( $12^\circ\text{C}$ ) Cu-holder providing cooling from all 4 lateral sides. The compact (microchip-type) laser cavity consisted of a plane pump mirror coated for HT at 780-1000 nm and for HR at 1800-2100 nm, and a flat output coupler (OC) having a transmission  $T_{OC}$  of 1.5%...20% at the laser wavelength. The pump source was a fiber-coupled (fiber core diameter:  $200 \mu\text{m}$ , N.A. = 0.22) CW AlGaAs laser diode emitting up to 14 W of unpolarized light at 802 nm (FWHM = 3 nm). Its output was collimated and focused into the active element through the PM by a lens assembly (1:1 imaging ratio, focal length  $f = 30 \text{ mm}$ ). The radius of the pump beam waist  $w_p$  and its Rayleigh length  $2z_R$  were  $100 \mu\text{m}$  and  $1.73 \text{ mm}$  ( $M^2 \sim 86$ ), respectively. The pump absorption (double-pass) under lasing conditions was 62%.

The input-output dependences and typical laser emission spectra are shown in Figs. 5(b) and 5(c). The laser output was linearly polarized ( $\sigma$ ); the polarization was naturally selected by the gain anisotropy. The maximum output power reached 1.82 W at 1945 nm with a slope efficiency  $\eta$  of 28.4% (with respect to the absorbed pump power  $P_{\text{abs}}$ ) for  $T_{OC} = 9\%$ . The laser threshold was at  $P_{\text{abs}} = 0.80 \text{ W}$ . With the increase of  $T_{OC}$ , the emission wavelength blue-shifted from 1955 nm to 1931 nm, Fig. 5(c), in agreement with the gain spectra, Fig. 4(b). We have also studied laser performance of a similar *c*-cut crystal ( $t = 3.4 \text{ mm}$ ). The laser generated 1.22 W at 1944 nm with higher  $\eta = 34.4\%$  (for  $T_{OC} = 9\%$ ). The laser emission was unpolarized. The laser operation in the plano-plano cavity indicates positive thermal lens for both *a*-cut and *c*-cut crystals.

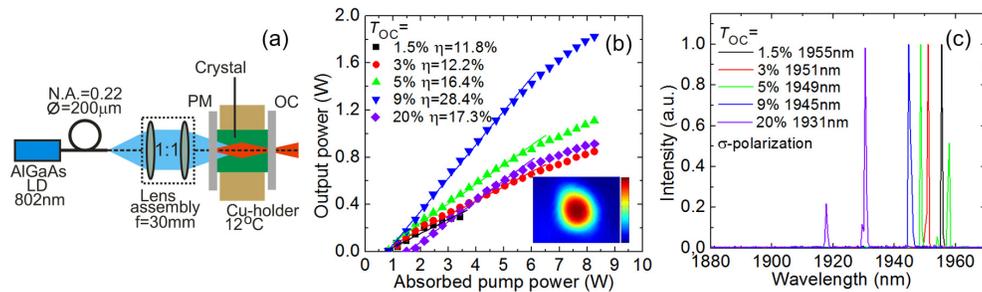


Fig. 5. (a-c) CW diode-pumped Tm:Ca(Gd,Lu)AlO<sub>4</sub> laser: (a) laser set-up: LD – laser diode, PM – pump mirror, OC – output coupler; (a) input-output dependences,  $\eta$  – slope efficiency, *inset* – measured spatial profile of the laser beam for  $T_{OC} = 9\%$  and maximum  $P_{\text{abs}}$ ; (b) laser emission spectra measured at the maximum  $P_{\text{abs}}$ . The crystal is *a*-cut and the laser polarization is  $\sigma$ .

An *a*-cut 1.8 at.% Tm:CaGdAlO<sub>4</sub> crystal ( $t = 3.0 \text{ mm}$ ) was inserted in the same laser set-up for comparison yielding 1.16 W at 1944 nm with  $\eta = 31.8\%$ . The laser emission was  $\sigma$ -polarized. Thus, the laser performance of Lu<sup>3+</sup>-codoped crystals is similar to that of Tm:CaGdAlO<sub>4</sub>.

The potential of Tm:Ca(Gd,Lu)AlO<sub>4</sub> for ML laser operation was revealed by studying its wavelength tuning performance in an X-shaped laser cavity, Fig. 6(a). The active element (*a*-cut,  $t = 6.0 \text{ mm}$ , aperture:  $3.0 \times 3.0 \text{ mm}^2$ ) was antireflection (AR) coated for pump and laser

wavelength. It was inserted in the cavity at normal incidence between two folding curved mirrors M1 and M2 (radius of curvature, RoC = 100 mm). M3 was a HR mirror and the transmission of the flat broadband OC was 0.5%...10%. The pump source was a CW Ti:Sapphire laser delivering >3 W at 798 nm ( $\sigma$ -polarization in the crystal). The pump was focused by a spherical lens ( $f = 70$  mm) to a spot with  $w_p = 30$   $\mu\text{m}$ . The pump absorption under lasing conditions measured near the laser threshold was 80.5% (single-pass).

Without any wavelength-selective element in the cavity, the free-running laser generated a maximum output power of 1.18 W at 1942 nm with a maximum  $\eta$  of 51.4% for  $\sigma$ -polarization ( $a$ -cut crystal,  $T_{OC} = 10\%$ ). The laser threshold was at  $P_{abs} = 0.13$  W. This high slope efficiency value indicates the effect of cross-relaxation for adjacent  $\text{Tm}^{3+}$  ions,  ${}^3\text{H}_4 + {}^3\text{H}_6 \rightarrow {}^3\text{F}_4 + {}^3\text{F}_4$ , increasing the pump quantum efficiency above unity. Indeed, the Stokes efficiency for this laser  $\eta_{St}$  is 40.8%.

For the wavelength-tuning experiment, a Lyot filter was inserted in the laser cavity near the OC, Fig. 6(a). It was a 3.2-mm-thick quartz plate with the optical axis at  $60^\circ$  to the surface. Using the  $a$ -cut  $\text{Tm:Ca}(\text{Gd,Lu})\text{AlO}_4$  crystal, the laser wavelength was continuously tuned from 1836 to 2083 nm (tuning range  $\Delta\lambda = 247$  nm,  $T_{OC} = 0.5\%$ ) and from 1827 to 2071 nm ( $\Delta\lambda = 244$  nm,  $T_{OC} = 1.5\%$ ), Fig. 6(b). A similar experiment was also performed for the  $c$ -cut crystal resulting in a tuning range from 1814 to 2072 nm ( $\Delta\lambda = 258$  nm,  $T_{OC} = 0.5\%$ ). The laser polarization was  $\sigma$  in all cases.

For achieving fs pulses in ML thulium-doped lasers, it is desirable that the  $\text{Tm}^{3+}$  emission extends above 2  $\mu\text{m}$  avoiding unwanted atmosphere water vapor absorption. This condition is satisfied in the  $\text{Tm:Ca}(\text{Gd,Lu})\text{AlO}_4$  laser. The introduction of  $\text{Lu}^{3+}$  ions provided additional broadening of the emission spectra, Fig. 3(a). This effect is confirmed in the tuning curves. Indeed, for the  $\text{Tm:CaGdAlO}_4$  ( $a$ -cut), laser spectral tuning was limited to 2065 nm.

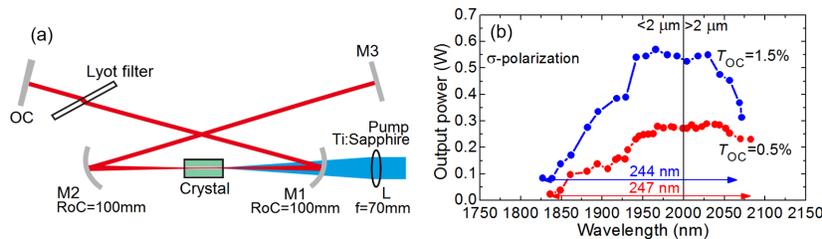


Fig. 6. (a,b) Wavelength tuning of the  $\text{Tm:Ca}(\text{Gd,Lu})\text{AlO}_4$  laser ( $a$ -cut crystal): (a) laser setup: M1 and M2 – folding mirrors, M3 – HR mirror, OC – output coupler, L – lens; (b) wavelength tuning curves ( $P_{abs} = 2.6$  W, laser polarization:  $\sigma$ ).

## 5. Conclusion

To conclude, we report on the growth, structural and spectroscopic characterization, and first CW and wavelength-tunable laser operation of a novel “mixed” disordered calcium aluminate crystal,  $\text{Tm:Ca}(\text{Gd,Lu})\text{AlO}_4$ . The introduction of  $\text{Lu}^{3+}$  preserves the tetragonal structure and induces additional spectral broadening for the  $\text{Tm}^{3+}$  absorption and emission bands with respect to  $\text{Tm:CaGdAlO}_4$ . As a consequence, it is attractive for fs ML lasers at  $\sim 2$   $\mu\text{m}$ . Further work will focus on the codoping of these crystals with  $\text{Tm}^{3+}$  and  $\text{Ho}^{3+}$  ions, to shift the emission range further beyond 2  $\mu\text{m}$  (due to the Ho ions), where water vapor absorption has a minor effect on broadband laser generation.

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