

# Molecular Characteristics of a Mixed-Valence Polyoxovanadate $\{V^{IV/V}_{18}O_{42}\}$ in Solution and at the Liquid–Surface Interface

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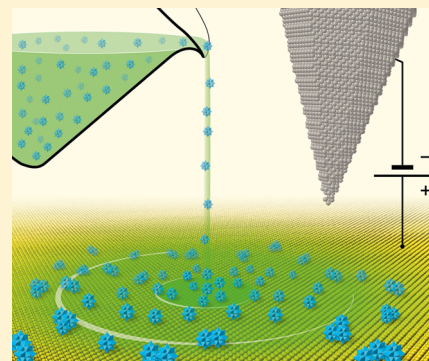
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## Supporting Information

**ABSTRACT:** The understanding of the molecular state of vanadium-oxo clusters (polyoxovanadates, POVs) in solution and on surface is a key to their target application in catalysis as well as molecular electronics and spintronics. We here report the results of a combined experimental and computational study of the behavior of nucleophilic polyoxoanions  $[V^{IV}_xV^V_yO_{42}(I)]^{5-}$  charged balanced by  $Et_4N^+$  in water, a one-phase organic solution of *N,N*-dimethylformamid (DMF) or acetonitrile (MeCN), a mixed solution of MeCN–water, and at the hybrid liquid–surface interface. The molecular characteristics of the compound  $(NEt_4)_5[V_{18}O_{42}(I)]$  (1) in the given environments were studied by microspectroscopic, electrochemical, scattering, and molecular mechanics methods. Contrary to the situation in pure water, where we observe great agglomeration with a number of intercalated  $H_2O$  molecules between POVs that are surrounded by the  $Et_4N^+$  ions, no and only minor agglomeration of redox-active POVs in an unprecedented cation-mediated fashion was detected in pure DMF and MeCN, respectively. An inclusion of 1% water in the MeCN solution does not have an effect significant enough to reinforce agglomeration; however, this leads to the POV...POV interface characterized by the presence of the  $Et_4N^+$  ions and a small number of  $H_2O$  molecules. Water amounts of  $\geq 5\%$  trigger the formation of higher oligomers. The deposition of compound 1 from MeCN onto an Au(111) surface affords nearly round-shaped particles ( $\sim 10$  nm). The use of DMF instead of MeCN results in bigger, irregularly shaped particles ( $\sim 30$  nm). This change of solvent gives rise to more extensive intermolecular interactions between polyoxoanions and their counterions as well as weaker binding of ion-pairing induced agglomerates to the metallic substrate. Lower concentration of adsorbed molecules leads to a submonolayer coverage and an accompanied change of the POV's redox state, whereas their higher concentration results in a multilayer coverage that offers the pristine mixed-valence structure of the polyoxoanion. Our study provides first important insights into the reactivity peculiarities of this redox-responsive material class on a solid support.



## INTRODUCTION

Owing to the great potential to exploit their inherent properties in a unique combination (structure/charge/redox state/magnetic state/resistivity), reduced polyoxovanadate (POV) single molecules attract significant attention among experimentalists and theoreticians working in the field of polyoxometalates<sup>1–6</sup> (metal-oxo clusters, abbreviated as POMs), the latter find a wide range of applications from catalysis<sup>7</sup> to nano-electronics.<sup>8,9</sup> The “fully-reduced” and mixed-valence POV structures are nowadays the subject of studies in biomedicine<sup>10</sup> as well as toward molecular quantum cellular automata,<sup>11,12</sup> “nanoscale quantum magnets”, spin qubits, and spintronic devices.<sup>13–16</sup> Their target application in supported-POV heterogeneous catalysis<sup>17–20</sup> and in technologically relevant

charge transport and optical mechanisms through individual, redox-responsive magnetic vanadium-oxo clusters,<sup>21</sup> however, requires control over the underlying characteristics and operation of these commonly negatively charged molecules in solution and on the substrate surface upon molecular deposition from the liquid phase.

Although the synthesis of POMs frequently occurs in an aqueous medium at controlled pH, in specific cases their direct microspectroscopic investigation<sup>22–31</sup> may mandate the use of an organic solvent. Indeed, an in-depth analysis of the

Received: March 6, 2017

Revised: April 12, 2017

Published: April 24, 2017



hypersensitive relationships of nucleophilic POM structures with their charge-balancing cations<sup>32</sup> (quaternary ammonium cations, ionic liquid cations, and/or alkali metal cations) and solvent molecules may substantially facilitate effective and efficient implementation of POMs in devices of interest. The POM-counterion and POM-solvent interactions have far-reaching consequences for the structure-property relations and reactivity from the perspective of applied materials and surface science.

While ion-pairing<sup>33</sup> between POM anions<sup>34</sup> and their counterions<sup>35</sup> as well as defined aggregation<sup>36</sup> of individual POMs into hierarchical self-assemblies,<sup>37</sup> micelles or vesicles<sup>38–41</sup> have stood in the focus of intense experimental and computational investigations over the last 10 years, the details of POM-counterion/solvent interactions remain almost unexplored<sup>42,43</sup> for POVs. Additionally, the correlation between the aggregation or agglomeration behavior of the particles in solution and after adsorption on a metallic surface represents a new aspect of research in POM chemistry, which is crucial for the future integration of POMs in catalytic systems and the molecule-based electronic and spintronic devices. With the aim of providing insights into the behavior of a POV bulk sample in solution and on metallic surfaces, we probed the tetraethylammonium (TEA = Et<sub>4</sub>N<sup>+</sup>) salt of a mixed-valence, host-guest<sup>44</sup> vanadium-oxo cluster [V<sup>IV</sup><sub>10</sub>V<sup>V</sup><sub>8</sub>O<sub>42</sub>(I)]<sup>5-</sup> (hereafter referred to as V<sub>18</sub>) synthesized<sup>45</sup> by Müller and co-workers. Hereby, we set the focus on the agglomeration properties of this polyoxoanion in solvents with different polarity and after adsorption on the well-studied Au(111) model surface.

The interest in [V<sup>IV</sup><sub>10</sub>V<sup>V</sup><sub>8</sub>O<sub>42</sub>(I)]<sup>5-</sup> containing reduced vanadyl {V<sup>IV</sup>O}<sup>2+</sup> groups with isotropic spin 1/2 is triggered by its easy synthetic accessibility, thermodynamic stability, and potentially facile switching opportunities between V<sup>IV</sup>/V<sup>V</sup> ratios as indicated by an isostructural series of the host-guest polyoxoanions exhibiting a spherical {V<sub>18</sub>O<sub>42</sub>} cluster shell (18:0 (“fully reduced”),<sup>46</sup> 16:2, and 10:8 (mixed valence)<sup>45</sup>). Moreover, the structural motif of V<sub>18</sub> is archetypal for numerous semimetal-substituted POV materials<sup>47</sup> that take part in catalytic processes<sup>48</sup> and are characterized by intriguing magnetic properties.<sup>47</sup> The stability, redox activity, and the molecular state of the title (NEt<sub>4</sub>)<sub>5</sub>[V<sub>18</sub>O<sub>42</sub>(I)] compound (**1**) in aqueous and organic solutions were studied by ultraviolet-visible (UV/vis) spectroscopy, cyclic voltammetry (CV), small-angle X-ray scattering<sup>49,50</sup> (SAXS), and molecular dynamics (MD) simulations. The molecular characteristics of this compound deposited from solution on the Au(111) surface were analyzed by scanning tunneling microscopy (STM) and X-ray photoelectron spectroscopy (XPS).

## EXPERIMENTAL AND COMPUTATIONAL METHODS

**Synthesis of Compound 1.** The synthesis of compound **1** was performed according to the published method by Müller et al.<sup>45</sup> The molecular composition and the purity of crystalline compound **1** were confirmed by standard elemental analysis, powder X-ray diffraction (Figure S1), and infrared spectroscopy (IR) (see the Supporting Information).

**Small Angle X-ray Scattering (SAXS).** SAXS experiments were performed on all samples (see Table 2) using GALAXI<sup>51</sup> at the Jülich Centre for Neutron Science, Forschungszentrum Jülich. The wavelength of X-ray radiation from a liquid GaInSn target was 0.134 nm. Detector distances of approximately 3.5 and 0.8 m were used to yield a *Q*-range from 0.05 to 6 nm<sup>-1</sup>. *Q*

is the scattering vector and is defined as  $(4\pi/\lambda) \sin(\vartheta/2)$ ,<sup>122</sup> where  $\vartheta$  is the scattering angle. The two-dimensional detector (Pilatus 1M) data were corrected for capillary scattering and scaled to absolute intensity [cm<sup>-1</sup>] through the scattering of a secondary polymer standard film (Fluoro-ethylene-propylene, Dupont) previously calibrated to water at the ID02 SAXS diffractometer of the ESRF, Grenoble, France. Before this, the data were azimuthally averaged using FIT-2D of ESRF. The samples were sealed in quartz capillaries of 1.5 mm inner diameter, placed in the vacuum chamber, and measured at ambient temperature. Solvents were treated in the same way and were subtracted from the data, weighted with their volume fraction. Since the *Q*-range overlap of both detector distances coincided in the peaked region (vide infra), for the subsequent fitting only the shorter detector distance was used without loss of information or generality.

**Molecular Dynamics (MD) Simulations.** MD simulations were carried out using GROMACS 4.5.4 software<sup>52–54</sup> and the AMBER 99 force field,<sup>55</sup> which has been successfully employed by Chaumont and Wipff<sup>56–58</sup> to study the aggregation of polyoxotungstates in solution. Parameters for the topology of the [V<sup>IV</sup><sub>10</sub>V<sup>V</sup><sub>8</sub>O<sub>42</sub>(I)]<sup>5-</sup> polyoxoanion and the Et<sub>4</sub>N<sup>+</sup> counterion, namely, V<sub>18</sub> and TEA, respectively, were obtained following the procedure developed by Poblet and co-workers.<sup>59,60</sup> The structure of V<sub>18</sub> was fully optimized using the Gaussian09 package<sup>61</sup> at the DFT level (BP86 functional)<sup>62</sup> and the LANL2DZ basis set.<sup>63</sup> For the organic TEA we used a Pople-type double- $\zeta$  basis set supplemented with polarization functions.<sup>64–66</sup> In addition, solvent effects were included during the geometry optimization using the dielectric IEF-PCM model.<sup>67</sup> CHelpG atomic charges were obtained by means of DFT single-point calculations in vacuum on the optimized structures at the same level of theory. Lennard-Jones parameters for O atoms belonging to the metal-oxo skeleton were taken from ref 59, and those for V atoms were taken from the UFF force field.<sup>68</sup> TIP3P water model<sup>69</sup> was used to represent solvent water molecules. MeCN and DMF were described by the models provided by van der Spoel et al.<sup>70</sup>

For the MD simulations, 10 V<sub>18</sub> polyoxoanions were randomly embedded in a cubic solvent box in which 50 TEA counterions were inserted at random positions to preserve the electroneutrality of the whole system. For simulations in water, the box size was (119.0 Å)<sup>3</sup> (54459 water molecules). For those in pure MeCN and DMF, the box size was (118.4 Å)<sup>3</sup> (18151 and 12369 solvent molecules, respectively). In calculations on wet MeCN including 1% volume of water, the simulation box contained 17716 MeCN and 471 water molecules, and on those including 5% volume of water, the box contained 16382 MeCN and 2723 water molecules. In all cases the POV concentration ranges from 0.010 to 0.009 mol·L<sup>-1</sup>. All simulations were performed with 3D-periodic boundary conditions using an atom cutoff of 14 Å for 1–4 van der Waals and of 10 Å for 1–4 Coulombic interactions and corrected for long-range electrostatics by using the particle-particle mesh Ewald (PME) summation method.<sup>71</sup> Five independent 5 ns dynamics were performed at 300 K starting with random velocities. The temperature was controlled by coupling the system to a thermal bath using the Berendsen algorithm<sup>72</sup> with a relaxation time of 0.1 ps to keep the canonical conditions (NVT) along the whole simulation. In simulations within an isothermal-isobaric (NPT) ensemble, the system was also coupled to a barostat using the Berendsen algorithm<sup>72</sup> with a relaxation time of 0.1 ps. Newton equations of motion were

185 integrated using the leapfrog algorithm,<sup>73</sup> and a time step of 1  
186 fs. Bonds involving H atoms were restrained by means of the  
187 LINCS algorithm.<sup>74</sup> Before the production runs, all the systems  
188 were equilibrated by 1000 steps of energy minimization  
189 followed by a 250 ps simulation at constant volume (NVT)  
190 fixing the solute. Then, a 250 ps NVT run was carried out with  
191 the solute relaxed, a 500 ps run at constant temperature (300  
192 K) and pressure (1 atm) (NPT), and finally a last 250 ps run  
193 (NVT).

194 **Surface Sample Preparation.** The (111) oriented gold  
195 single crystal used for the adsorption experiments was  
196 purchased by MaTeck, Germany. The surface preparation was  
197 done in a separate ultrahigh vacuum (UHV) chamber by  
198 repeated cycles of neon ion sputtering ( $I \approx 3 \mu\text{A}$  at  $p(\text{Ne}) = 2$   
199  $\times 10^{-6}$  mbar) and subsequent annealing at 1000 K. The so-  
200 cleaned substrate was then transferred to ambient conditions  
201 for coating with 1  $\mu\text{L}$  of a freshly prepared POV solution and  
202 afterward quickly retransferred into UHV. Dry MeCN and  
203 DMF were used as solvents. In both solvents a complete  
204 dissolving of compound **1** was observed, so that the given  
205 concentrations were directly calculated by the used quantity of  
206 the solid compound.

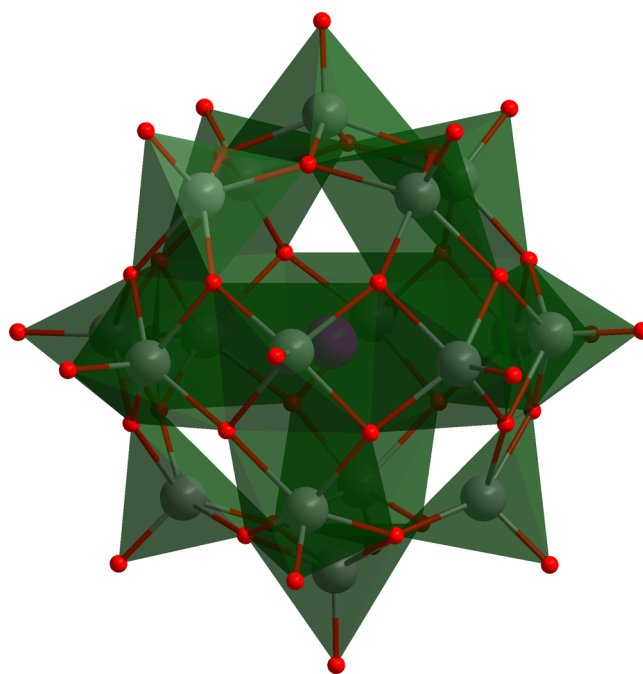
207 **Scanning Tunneling Microscopy (STM).** STM experi-  
208 ments were conducted under UHV conditions at room  
209 temperature in an Omicron VT-SPM. The base pressure inside  
210 the stainless steel vessel during operation was  $5 \times 10^{-10}$  mbar.  
211 For STM imaging, mechanically cut Pt–Ir tips were used. All  
212 images were measured in constant current mode. The SPIP  
213 5.1.11 software from Image Metropolis was used for STM data  
214 analysis.

215 **X-ray Photoelectron Spectroscopy (XPS).** XPS measure-  
216 ments were performed with a PHI 5000 Versa Probe (Physical  
217 Electronics Inc., USA) with Al  $K\alpha$  X-ray illumination, a pass  
218 energy of 58.7 eV, and a photoemission angle of  $55^\circ$  using  
219 electron neutralization. The stainless steel vessel had a base  
220 pressure of  $1 \times 10^{-9}$  mbar. For binding energy reference, the C  
221 1s peak at 285 eV was used. The spectra were fitted to a  
222 convolution of a Lorentz profile and a Gaussian profile after  
223 subtraction of a Shirley background.

## 224 ■ RESULTS AND DISCUSSION

225 **Structure and Redox Properties of  $V_{18}$ .** The structure<sup>45</sup>  
226 of the herein studied host–guest  $V_{18}$  polyoxoanion is illustrated  
227 in Figure 1. It consists of the 18 self-assembled, edge-sharing  
228 square-pyramidal  $\{\text{O} = \text{VO}_4\}$  units shaping a spherical  
229 vanadium-oxo shell with a diameter of ca. 11 Å, which  
230 encapsulates an iodide ion ( $\text{I}^-$ ) in the central void. The high  
231 negative charge of this POV component in compound **1** is  
232 counterbalanced by five TEA cations.

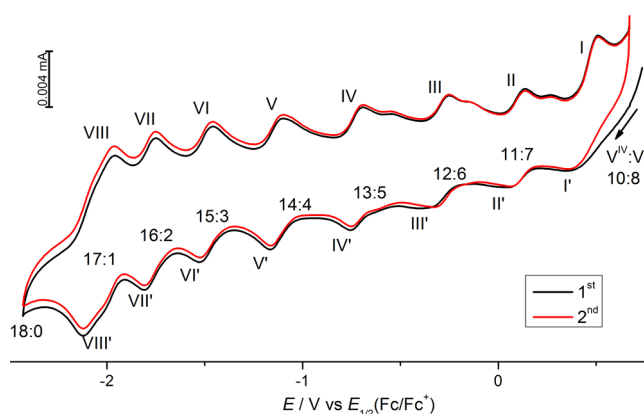
233 Because compound **1** shows good solubility in MeCN and  
234 DMF we spectroscopically tested the stability of  $V_{18}$  in these  
235 organic solvents and additionally under slightly harsher  
236 conditions, namely, in buffered aqueous solutions of HOAc/  
237 KOAc ( $\text{OAc}^- = \text{acetate}$ ) in the range pH 3–6. A typical broad  
238 intervalence ( $V^{\text{IV}}/V^{\text{V}}$ ) charge transfer (IVCT) band<sup>75,76</sup> at ca.  
239  $\lambda_{\text{max}} = 920$  nm has been observed in the UV/vis spectra of **1** in  
240 the specified solutions, thus demonstrating the mixed-valence  
241 character of  $V_{18}$ . Exception is the situation in the HOAc/KOAc  
242 medium at pH lower than 5 where the IVCT band does not  
243 appear, which indicates a degradation of the title POV (Figures  
244 S2–S4 in the Supporting Information). A comparative analysis  
245 between the solid-state IR spectrum of polycrystalline **1** and  
246 those spectra obtained after evaporation of the organic



**Figure 1.** Combined ball-and-stick and polyhedral representation of a molecular structure of the spherical  $V_{18}$  polyoxoanion in compound **1**. Color code: V = gray spheres; O = red spheres;  $\{\text{VO}_4\}$  = transparent green square pyramids.

247 solutions of **1** under an ambient atmosphere (Figure S5 in 247  
the Supporting Information) suggests that the solid compound 248  
maintains its structural integrity after solvation in MeCN or 249  
DMF. In addition, the solid-state XPS spectrum of **1**, discussed 250  
below, confirms the mixed-valence form of the  $V_{18}$  polyoxo- 251  
anion. 252

253 As a next step, we assessed the redox properties of the mixed- 253  
valence  $V_{18}$  polyoxoanion by CV measurements in a 0.1 M 254  
TEAPF<sub>6</sub>–MeCN solution ( $T = 25^\circ\text{C}$ ) for  $dE/dt = 20$ – $200$  255  
 $\text{mV}\cdot\text{s}^{-1}$  in the range of  $E = -2.43$ – $0.67$  V using the half-wave 256  
potential ( $E_{1/2}(\text{Fc}/\text{Fc}^+) = +0.380$  V vs saturated calomel 257  
electrode)<sup>77</sup> of ferrocene/ferrocenium ( $\text{Fc}/\text{Fc}^+$ ) as an external 258  
standard (see Figures S6–S8 in the Supporting Information). 259  
The main results of this electrochemical analysis are shown in 260  
Figure 2 and summarized in Table 1. 261



**Figure 2.** Cyclic voltammogram (1st and 2nd cycles) of compound **1** (0.3 mM) in a 0.1 M TEAPF<sub>6</sub>–MeCN solution, with a scan rate of  $50 \text{ mV}\cdot\text{s}^{-1}$  vs.  $\text{Fc}/\text{Fc}^+$ .

**Table 1.** Data from Cyclic Voltammograms Presented in Figure 2 and Figures S6–S8 in the Supporting Information

$dE/dt$ [mV·s <sup>-1</sup> ]	$E_{1/2}$ [V] <sup>a</sup>		$\Delta E_{\text{ox/red}}$ [V]		$i_{\text{pc}}/i_{\text{pa}}$
	50	100	50	100	
I/I'	0.45	0.37	0.12	0.15	
II/II'	0.11	0.03	0.06	0.06	1.0
III/III'	-0.28	-0.35	0.06	0.06	1.0
IV/IV'	-0.72	-0.79	0.06	0.06	0.9
V/V'	-1.12	-1.20	0.06	0.06	0.9
VI/VI'	-1.48	-1.56	0.06	0.06	0.9
VII/VII'	-1.78	-1.85	0.06	0.06	1.0
VIII/VIII'	-2.04	-2.11	0.15	0.16	

<sup>a</sup>Solution of 0.3 mM in 0.1 M TEAPF<sub>6</sub>-MeCN vs E<sub>1/2</sub>(Fc/Fc<sup>+</sup>).

The first and second cycles depicted in the cyclic voltammogram (Figure 2) reveal a Coulomb staircase of eight reduction waves and the corresponding oxidation waves for  $dE/dt = 50 \text{ mV}\cdot\text{s}^{-1}$ , which reflect the maximum possible 8-fold one-electron reduction of V<sub>18</sub> (that is {V<sup>IV</sup><sub>10</sub>V<sup>V</sup><sub>8</sub>} → {V<sup>IV</sup><sub>18</sub>V<sup>V</sup><sub>0</sub>}) and the following reoxidation to the initial electron population, respectively. For  $E > 0.7 \text{ V}$  the irreversible decomposition of compound **1** on the electrode surface is observed. This is in agreement with the fact that no isostructural {V<sub>18</sub>O<sub>42</sub>(X)} complexes oxidized greater than V<sub>18</sub> with V<sup>IV</sup>/V<sup>V</sup> = 10:8 could be obtained so far (see {V<sup>IV</sup><sub>18-n</sub>V<sup>V</sup><sub>n</sub>(X)})<sup>45</sup> where  $n = 0, 2, \text{ or } 8$  and X = anionic or neutral guest).

For waves II, III, and IV distinct shouldering effects in the anodic region are identifiable. These effects arose probably due to minor ion pairing hindering the oxidation of V<sub>18</sub>. The peak-to-peak distances ( $\Delta E_{\text{ox/red}}$  in Table 1) for the redox couples II/II'–VII/VII' at two different scan rates (50 and 100 mV·s<sup>-1</sup>) fit with the ideal value of  $\Delta E_{\text{ox/red}} = 59 \text{ mV}$  at 25 °C<sup>78</sup> for reversible one-electron transfers. The slight deviation from this value can be attributed to some uncompensated resistance.<sup>9</sup> This is also in accordance with the peak-current ratios ( $i_{\text{pc}}/i_{\text{pa}}$ ) of II/II'–VII/VII', which agree closely with the ideal value of 1;<sup>78</sup> minor differences may arise in consequence of kinetic limitation (refer to a transfer coefficient  $\alpha > 0.5$ ).<sup>78</sup> For the redox couples I/I' and VIII/VIII' the peak–current ratio could not be determined exactly. Furthermore, I/I' and VIII/VIII' exhibit much higher peak separations, which can be associated with an electrochemically irreversible but chemically reversible behavior or with a possible multiple electron transfer caused by adsorption on the cathode (VIII/VIII') or convolution/superposition effects as a consequence of the irreversible decomposition on the anodic electrode surface (I/I') at higher potentials. For all redox couples the peak height is proportional to the square root of the scan rate (see the Supporting Information), confirming an ideal diffusion-controlled behavior; although, a slight shift of E<sub>1/2</sub> to lower potentials is observed (see Table 1).

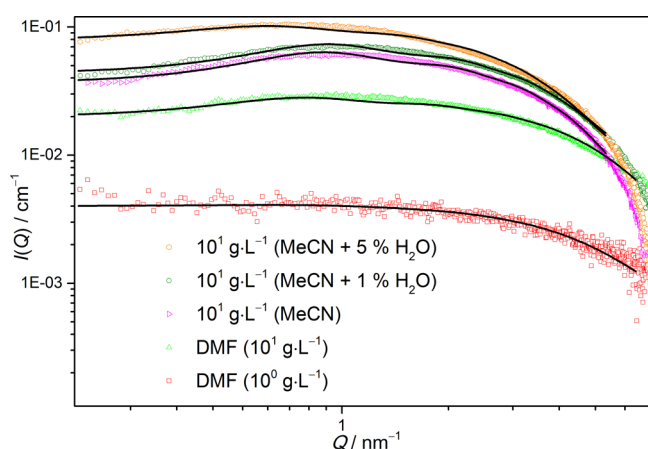
In comparison with CV measurements in MeCN, those in H<sub>2</sub>O (conditions: 0.1 M KCl or LiClO<sub>4</sub> electrolyte; Ag/AgCl reference electrode; glassy carbon working electrode; platinum wire counter electrode) did not show any discrete redox waves. This can have several possible causes such as strong agglomeration or protonation effects.

It is also noteworthy that the determination of diffusion coefficients and hydrodynamic radii for compound **1** in MeCN via standard dynamic light scattering techniques has failed, in particular because of the absorption of laser light ( $\lambda = 632 \text{ nm}$ ) lying in the broad IVCT region of V<sub>18</sub> (see the Supporting

Information). We therefore sought further details of the solution behavior of compound **1** by SAXS.

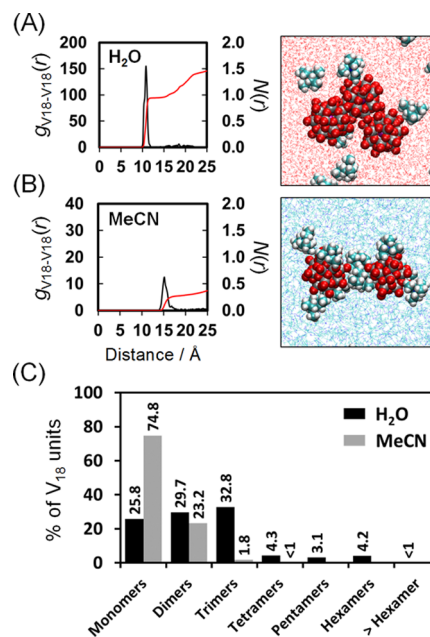
**SAXS Measurements.** The SAXS data covered length scales up to a few tens of nanometers. Due to the peaked structure they were interpreted in terms of intraparticle (form factor  $P(Q)$ ) and interparticle (structure factor  $S(Q)$ ) correlations. For the shape of the V<sub>18</sub> polyoxoanions a rotationally averaged spherical geometry (see Figure 1) was assumed. The scattering length density was calculated from the electron density of V<sub>18</sub> by applying the gross formula in which the bulk density of 1.91 g·cm<sup>-3</sup> was used. Tabulated densities for the respective calculation for MeCN and DMF were used. With the squared difference of scattering length density between these two components, that is the contrast  $\Delta\rho^2$  and the volume fraction in the dilute solution of  $\sim 1 \text{ mg}\cdot\text{mL}^{-1}$  in DMF, both the Q-dependence yielding the radius  $R_p$  of the spherical entity as well as the volume of the scattering particle  $V_p$  from the extrapolated experimental forward scattering  $I(Q = 0)$  are in accordance within better than 10% with a monomeric V<sub>18</sub> conformation. This agreement can be judged excellent given the experimental uncertainties in concentrations, densities, absolute calibration, and capillary scattering subtractions. At this concentration, the data can thus be interpreted directly as  $P(Q)$ , the form factor of the V<sub>18</sub> polyoxoanions. We note that the counteraction TEA cannot be distinguished from either DMF or water and thus the pure anionic vanadium-oxo cluster is determined. In MeCN a small contribution of the shell could be expected, however, leading to a larger effective particle radius. At a 10-fold higher concentration in DMF as well as in mixed solvents of MeCN and water, the SAXS data show an additional broad peaked intensity pattern around  $Q \approx 0.9 \text{ nm}^{-1}$  indicative for a Bragg-like peak or from a liquid-like ordering, which must be described by an appropriate structure factor  $S(Q)$ . The latter reduces to 1 if the vanadium-oxo clusters are dilute and do not experience the presence of neighboring clusters. Although they are charged species but are complexed with counterions, an effective hard-sphere potential is assumed here for the interaction.<sup>79</sup> This potential has been shown previously<sup>80</sup> to be a good approximation for both soft matter and hard matter systems. Due to its analytical form it provides an ideal way to parametrize the scattering data. Thus, the intensity curves  $I(Q)$  were fitted with  $I(Q) = \phi\Delta\rho^2V_pP(Q,R_p)S(Q,R_{\text{HS}},\eta)$ . Here, the volume  $V_p$  is calculated from the particle radius  $R_p$ , and the effective interaction hard sphere  $R_{\text{HS}}$  with an effective volume fraction  $\eta$  can be viewed as half the distance  $D$  between the centers-of-mass of interacting V<sub>18</sub> polyoxoanions. This  $D$  can be roughly estimated already from the relation  $D \approx 2\pi/Q_{\text{peak}}$ . The formal expressions for  $P(Q)$  and  $S(Q)$  along with some more details (Figure S9) are given in the Supporting Information. In the fitting an additional diffuse background contribution due to closeness of the amorphous halo was taken into account. The fits are shown in Figure 3. The agreement is good and the small remaining discrepancies are assigned to the monodisperse and simple-spherical assumption of V<sub>18</sub> as well as to details of the interaction potential. In reality this is presumably softer and should resemble more to a screened Coulomb dependence. However, because we are interested here merely in the question of monomeric or oligomeric vanadium-oxo clustering, the parametrization that focuses on the intercluster distance is sufficient.

The measured SAXS data show a weak dependence of the V<sub>18</sub> polyoxoanion sphere size from the used solvent. The  $R_p$



**Figure 3.** Absolute SAXS intensity data  $I(Q)$  (in  $\text{cm}^{-1}$ ) and corresponding fits to a spherical form factor and simulated hard-sphere interaction as a function of the scattering vector  $Q$  (in  $\text{nm}^{-1}$ ). The compound concentration of  $1 \text{ g}\cdot\text{L}^{-1}$  allows the form factor to be determined, whereas higher concentrations reveal non-negligible interactions. Due to the smallness of the particle, no oscillations of the form factor are observed, and the data are almost entirely in the Guinier regime.

results reflecting the behavior of  $V_{18}$  in the above-mentioned 400 solvents are shortly described as two important features (see 401 **Figure 4**): the radial distribution functions (RDFs) between 402 f4



**Figure 4.**  $V_{18}\cdots V_{18}$  RDFs between centers of mass (black curves) and typical snapshots of the agglomerate species resulted from simulations of  $V_{18}$  in  $\text{H}_2\text{O}$  (A) and pure MeCN (B). Red lines represent the integration of the RDFs ( $g$ ), that is the coordination number  $N(r)$ , which indicates the average number of neighboring molecules. Color code: V = gray spheres; O = red spheres; N = blue spheres; C = cyan spheres; H = white spheres. Summary of the % of POV units forming each different agglomeration state species averaged over 25 ns of MD trajectories in  $\text{H}_2\text{O}$  (black bars) or MeCN (gray bars) is given in (C).

$V_{18}\cdots V_{18}$  centers of mass, both in pure  $\text{H}_2\text{O}$  and MeCN, 403 accompanied by representative snapshots and the % of isolated 404 monomers and oligomers detected in the MD runs. In addition, 405 Table S1 in the **Supporting Information** summarizes the data 406 obtained from the RDF analysis of the studied systems. 407

**Simulations of  $V_{18}$  in Water.** Visual inspection of the MD 408 trajectories reveals, in addition to the presence of free 409 monomers, agglomeration of  $V_{18}$  units to form mainly dimers 410 and trimers, but also higher oligomers, in agreement with the 411 lack of discreteness observed in the redox CV waves of 412 compound **1** in water. Related to this,  $V_{18}\cdots V_{18}$  interactions are 413 much longer-lived than  $V_{18}\cdots\text{TEA}$  ones. In fact, the  $V_{18}\cdots\text{TEA}$  414 contacts that can be detected involve one or two TEA 415 molecules per POV at the most and exist only during short time 416 intervals, whereas most of the  $V_{18}\cdots V_{18}$  agglomerates formed 417 last for the rest of the simulation. The  $V_{18}\cdots V_{18}$  RDF, illustrated 418 in **Figure 4A**, displays a single sharp peak at ca.  $10.8 \text{ \AA}$  (roughly 419 the sum of two  $V_{18}$  radii) integrating to  $0.94 V_{18}$  units. This is 420 the average number of POVs surrounding another during the 421 computed trajectories. It must be pointed out that TEA cations 422 are partially hydrophobic, but not enough to form a 423 microphase, as has been observed for the bulkier  $n\text{-Bu}_4\text{N}^+$  424 cations.<sup>56</sup> Thus,  $V_{18}$  agglomeration occurs in a water-mediated 425 fashion, consisting of  $\text{POV}\cdots(\text{H}_2\text{O})_x\cdots\text{POV}$  patterns with a 426 number of intercalated water molecules between POV units 427 that are weakly surrounded by the TEA counteranions. This 428 pattern is also explained by the high  $\text{POV}\cdots\text{H}_2\text{O}$  affinity 429

373 value of  $0.63 \text{ nm}$  in MeCN in comparison to  $0.49 \text{ nm}$  in DMF 374 (**Table 2**) indicates a larger average particle size in the less polar

**Table 2.** SAXS Data and Modeling Shown in **Figure 3**<sup>a</sup>

solvent (concentration)	additional water content [%]	$R_p$ [nm]	$D$ [nm]
DMF ( $1 \times 10^0 \text{ g}\cdot\text{L}^{-1}$ )	0	0.49	6.10
DMF ( $1 \times 10^1 \text{ g}\cdot\text{L}^{-1}$ )	0	0.49	6.94
MeCN ( $1 \times 10^1 \text{ g}\cdot\text{L}^{-1}$ )	0	0.63	6.24
	5	0.72	8.44

<sup>a</sup>Error bars are of the order of 0.1%.

375 medium, which might be explained by either a small cation shell 376 contribution or a weaker charge compensation of MeCN. Ion 377 pairing effects might be favored due to a larger shielding effect 378 of the solvent. However, both values are in the range of mainly 379 monomeric polyoxoanions with only very few amount of 380 slightly larger particles.

381 Small water contents in the MeCN solvent in the range of 382 1% simulating typical wet solvents have no detectable influence 383 on the particle size (**Table 2**). However, water amounts of 5% 384 or greater result in a further increase of  $R_p$  indicating more and/ 385 or higher oligomers. This interpretation is also supported by 386 the significantly increased average interparticle distance  $D$ . The 387 formation of larger  $V_{18}$  agglomerates reduces the effective 388 amount of isolated scattering centers in the solution so that the 389 distance between them has to increase at the same time.

390 **MD Simulations.** The collected SAXS data prompted us to 391 conduct MD simulations to obtain more detailed information 392 about the characteristics of  $V_{18}$  in solution. The computations 393 were performed in pure water (solvent in the synthesis of  $V_{18}$ ), 394 pure MeCN, pure DMF, and wet MeCN containing 1% volume 395 of water, which represents the moisture that dry, high-purity 396 organic solvent could take during the experimental setups. To 397 explore the effect of including higher amounts of water in the 398 organic solvent we also performed simulations in wet MeCN 399 containing 5% volume of water. The main computational

430 (interaction energy of  $-1387 \text{ kJ}\cdot\text{mol}^{-1}$  per POV molecule, see  
431 Table 3). Indeed, the coordination number obtained from the

**Table 3. Interaction Energies<sup>a</sup> ( $E_{\text{int}}$ ) Obtained from MD Simulations per POV Unit and per TEA Unit, Respectively**

type of interaction	$E_{\text{int}}$ [ $\text{kJ}\cdot\text{mol}^{-1}$ ]
POV $\cdots$ H <sub>2</sub> O	-1387
POV $\cdots$ DMF	-735
POV $\cdots$ MeCN	-560
TEA $\cdots$ H <sub>2</sub> O	-88
TEA $\cdots$ DMF	-129
TEA $\cdots$ MeCN	-111

<sup>a</sup>Averaged noncovalent intermolecular energies calculated from values taken every 2 ps over a total simulation time of 25 ns.

432 integration of the first peak in  $V_{18}\cdots\text{TEA}$  RDF (displayed in  
433 Figure S10) quantifies an average of 0.33 TEA cations in close  
434 contact with each  $V_{18}$  unit. We expect the first shell of water  
435 molecules surrounding POVs not to be easily displaced to form  
436 agglomerates. In solution, TEA ions mediating POV $\cdots$ POV  
437 agglomerates were not observed at least for the present  
438 simulation time, but this possibility cannot be discarded  
439 completely. Results on longer simulation times are underway  
440 and will be published elsewhere.

441 From MD trajectories, agglomeration is accounted for when  
442 the distance between the centers of mass of  $V_{18}$  units is  $<13 \text{ \AA}$ .  
443 Thus, 25.8% of the  $V_{18}$  units in solution remain as monomers,  
444 while agglomerates are represented mainly by dimers (29.7%)  
445 and trimers (32.8%), but also higher oligomers (4.3% of POVs  
446 being tetramers and the remaining 7.4% pentamers and  
447 hexamers). These data are summarized in Figure 4C. In total,  
448 ca. 74% of the  $V_{18}$  polyoxoanions form some type of  
449 agglomerate along the simulation time at 0.01 M concentration,  
450 certainly subscribing the notable tendency of  $V_{18}$  to form  
451 agglomerates in water.

452 *Simulations of  $V_{18}$  in Pure MeCN and Pure DMF.* With  
453 either solvent, visual inspection of the MD trajectories suggests  
454 strongly mitigated agglomeration with respect to the results in  
455 aqueous solution. In MeCN we mainly observe the presence of  
456 monomolecular  $V_{18}$  species with the exception of sparse  
457 formation of dimers. In solvents other than water, the  
458 agglomeration process occurs in an unprecedented and intricate  
459 cation-mediated POV $\cdots$ (TEA)<sub>x</sub> $\cdots$ POV fashion. Consequently,  
460 the first peak in the corresponding  $V_{18}\cdots V_{18}$  RDF (Figure 4B)  
461 appears at a larger distance of ca. 15.1  $\text{\AA}$  and integrates only to  
462 0.25  $V_{18}$  (vs 0.94 in H<sub>2</sub>O).

463 Thus, larger POV $\cdots$ TEA ion pairing is expected in less polar  
464 organic solvents (see Figure S10 in the Supporting Information,  
465 which compares the  $V_{18}\cdots\text{TEA}$  RDFs computed in H<sub>2</sub>O, DMF,  
466 and MeCN). The basis for this is simply that the POV $\cdots$ solvent  
467 interaction is much weaker in organic solvents (see Table 3).  
468 Not only does the solvent polarity play a role but other sources  
469 of TEA $\cdots$ solvent affinity, such as dispersion or van der Waals  
470 interactions, can play a crucial role in limiting or aiding POV $\cdots$   
471 TEA pairing. In the present case, TEA has more affinity to  
472 DMF,  $-129 \text{ kJ}\cdot\text{mol}^{-1}$ , than to MeCN,  $-111 \text{ kJ}\cdot\text{mol}^{-1}$  (values  
473 shown in Table 3). This fact is ascribed to the role of the two  
474 methyl groups in DMF. We suggest, then, that the dipolar  
475 moment of the two organic solvents play a secondary role in  
476 the different behavior in solution since they are alike.

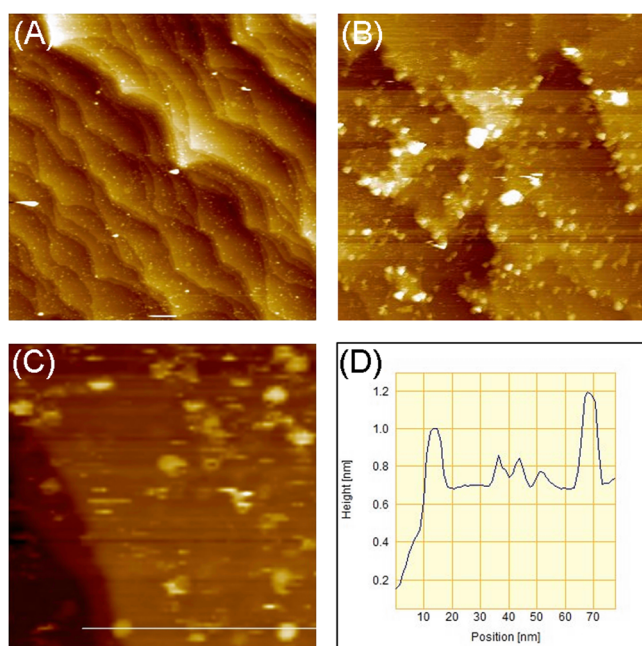
477 In MeCN, despite the considerable POV $\cdots$ TEA pairing  
478 computed (4.58 TEA per POV unit), the amount of  $V_{18}$

agglomeration is quite small ( $\sim 25\%$  vs  $\sim 74\%$  in water; see  
Figure 4C), that is, the majority of POVs in this solvent remain  
monomeric (74.8%). In DMF, the  $V_{18}\cdots V_{18}$  RDF (Figure S11  
in the Supporting Information) does not show any definite and  
unambiguous peak, suggesting a complete lack of agglomeration.  
In this case, the TEA $\cdots$ DMF and POV $\cdots$ DMF affinities  
( $-129$  and  $-735 \text{ kJ}\cdot\text{mol}^{-1}$ , respectively), stronger compared  
with the ones in MeCN, derive from a lack of the necessary  
POV $\cdots$ TEA pairing for agglomeration, which explains this  
result. These findings fully agree with the SAXS data and with  
the observation of the well-defined redox peaks obtained from  
CV experiments in MeCN solution, confirming that  $V_{18}$  chiefly  
remains monomolecular.

*Simulations of  $V_{18}$  in Wet MeCN.* Results in wet MeCN (1%  
volume of water) do not show clear differences in comparison  
with results in pure MeCN. A single peak at ca. 14.7  $\text{\AA}$  is  
observed in the  $V_{18}\cdots V_{18}$  RDF, integrating to 0.33  $V_{18}$  (Figure  
S12A in the Supporting Information). This indicates that  
moisture in the solvent does not increase the tendency of  $V_{18}$  to  
form agglomerates. We assume that our trajectories are long  
enough for this comparative study since several randomly  
chosen water molecules were able to explore a large part of the  
space (Figure S13 in the Supporting Information). Also, the  
 $V_{18}\cdots$ water RDF (Figure S12A in the Supporting Information)  
shows two peaks corresponding to the water molecules  
interacting with bridging and terminal atoms of the POV,  
respectively. This further demonstrates that water molecules  
reach the  $V_{18}$  units during the simulations. In the given case, the  
 $V_{18}\cdots V_{18}$  agglomerates are mediated by TEA cations as in the  
previous section but we can also identify some water-mediated  
interactions as shown in a typical snapshot of a POV $\cdots$ (TEA,  
H<sub>2</sub>O)<sub>x</sub> $\cdots$ POV dimer in Figure S12A (Supporting Information).  
This combination of TEA and water molecules at the POV $\cdots$   
POV interface in wet MeCN seems to be responsible for the  
shorter  $V_{18}\cdots V_{18}$  distance at the maximum of the peak in the  
RDF compared to that in pure MeCN. Nevertheless, when  
increasing the amount of water to 5%, we observe that the  
agglomeration slightly increases, as the peak in the correspond-  
ing  $V_{18}\cdots V_{18}$  RDF (Figure S12B in the Supporting Information)  
integrates to 0.41  $V_{18}$  (cf. 0.33 in simulations with 1% water).  
As the water molecules in the media tend to distribute around  
the POV (see  $V_{18}\cdots\text{H}_2\text{O}$  RDF in Figure S12B), the larger  
number of agglomerate species must be a consequence of a  
higher number of water-mediated  $V_{18}\cdots V_{18}$  contacts, as shown  
in Figure S14 (Supporting Information). This fact might  
explain the larger  $R_p$  observed for the 5% water-containing  
MeCN solution during SAXS measurements.

**Molecular Surface Deposition and Characterization by STM.** The use of combined redox, magnetic, and resistive properties of mixed-valence POVs in heterogeneous catalysis and in future single molecule-based nanoelectronic and nanospintronic devices mandates the controlled immobilization of these vanadium-oxo clusters onto a solid substrate. Therefore, we targeted the molecular deposition of  $V_{18}$  on the Au(111) surface from the liquid phase, which represents a first fundamental task in the technologically relevant multistep process. The substrate was chosen because of its well-investigated structure and easy preparation.<sup>81</sup> In order to stabilize the highly negatively charged  $V_{18}$  molecules without forming a huge solvent cloud we used the polar aprotic, dry solvents MeCN and DMF.

Figure 5A shows a representative STM image of the Au(111) surface, which was covered by a highly diluted  $V_{18}$ -MeCN



**Figure 5.** STM images of a submonolayer  $V_{18}$  covered Au(111) surface prepared by deposition (A) from a MeCN solution (concentration =  $0.9 \times 10^{-3} \text{ g}\cdot\text{L}^{-1}$ ;  $1000 \text{ nm} \times 1000 \text{ nm}$ ;  $U_B = 1.5 \text{ V}$ ,  $I_T = 0.5 \text{ nA}$ ) and (B) from a DMF solution (concentration =  $1.4 \times 10^{-3} \text{ g}\cdot\text{L}^{-1}$ ;  $900 \text{ nm} \times 900 \text{ nm}$ ;  $U_B = 1.5 \text{ V}$ ,  $I_T = 0.5 \text{ nA}$ ). Deposition from DMF solution results in bigger agglomerates and a preferred adsorption at the step edges of the substrate terraces. (C) High resolution scan of (A) ( $100 \text{ nm} \times 100 \text{ nm}$ ;  $U_B = 1.5 \text{ V}$ ,  $I_T = 0.5 \text{ nA}$ ). (D) Height profile from (C) along two presumable  $V_{18}$  polyoxoanions and some fragments.

542 solution with a concentration of compound **1** equal to  $0.9 \times$   
 543  $10^{-3} \text{ g}\cdot\text{L}^{-1}$ . Under these conditions a submonolayer coverage of  
 544 nearly round shaped particles with a typical lateral diameter of  
 545  $\approx 10 \text{ nm}$  and a height ranging from  $0.3$  to  $0.6 \text{ nm}$  is found on  
 546 most parts of the sample surface (see Figures 5C,D). Based on  
 547 their characteristic size and reproducible shape we suppose that  
 548 these particles may represent single  $V_{18}$  polyoxoanions, which  
 549 are agglomerated through their TEA counteranions. The  
 550 adsorbed particles show no obvious ordering on the surface. A  
 551 special preference of terrace step edges or other surface defects  
 552 is not detected in that case. Besides these characteristic particles  
 553 also a lot of smaller species with a less defined structure can be  
 554 observed on the surface. If these objects refer to  $V_{18}$  fragments  
 555 and/or solvent residues remains speculative.

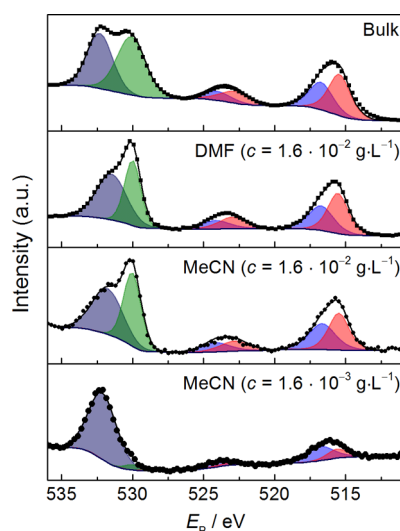
556 Using DMF with a slightly higher polarity (polarity index<sup>82</sup>  
 557 of  $6.4$  vs  $5.8$ ) and a much higher boiling point ( $153 \text{ }^\circ\text{C}$  vs  $81.6$   
 558  $^\circ\text{C}$ ) than MeCN changes the  $V_{18}$  appearance on the Au(111)  
 559 surface significantly. Figure 5B shows an STM image of a  
 560 submonolayer covered sample area deposited from a  $V_{18}$ -DMF  
 561 solution with a concentration of compound **1** equal to  $1.4 \times$   
 562  $10^{-3} \text{ g}\cdot\text{L}^{-1}$ . Compared to the situation after deposition from the  
 563 MeCN solution the observed particles are much larger with an  
 564 averaged lateral size of  $\sim 30 \text{ nm}$  and heights between  $1.5$  and  $6$   
 565  $\text{nm}$ . The particles have no regular shape anymore and are  
 566 obviously preferred located at the terrace step edges of the gold  
 567 surface. This indicates the formation of larger agglomerates of  
 568 several  $V_{18}$  polyoxoanions and their corresponding counterions  
 569 due to more extensive intermolecular interactions (that is ion  
 570 pairing), which, however, weakens the binding to the metallic  
 571 substrate.

Thus, it gets obvious that the used solvent has a strong  
 influence on the  $V_{18}$  adsorption behavior. We believe that the  
 significantly higher boiling temperature of DMF compared to  
 that of MeCN could be crucial for the larger agglomeration on  
 the gold surface. The longer evaporation time of the solvent  
 after the surface coating might lead to a higher surface mobility  
 during the preparation process, but also polarity effects cannot  
 be excluded although the SAXS measurements have shown an  
 opposite trend for the polyoxoanion agglomeration in solution.

It should be noted that STM measurements of the adsorbed  
 $V_{18}$  polyoxoanions turned out to be quite challenging. Probably  
 due to the high molecular charge of  $V_{18}$  interactions with the  
 STM tip appeared quite frequently. Scanning with negative bias  
 voltage (that is, with a positively charged tip) was hardly  
 possible due to attraction of the negatively charged  $V_{18}$   
 molecules. Another problem, which was encountered during  
 the measurements, was a somehow nonuniform  $V_{18}$  distribution  
 on the macroscopic scale. Adjacent to the above-mentioned and  
 mainly observed regions with submonolayer coverage also areas  
 without  $V_{18}$  and other regions with multilayer coverage were  
 found on the gold surface. We assume that the observed  
 nonuniform distribution may be the result of an inhomoge-  
 neous evaporation of the solvent droplet during the deposition  
 process since our SAXS measurements indicate no significant  
 $V_{18}$  agglomeration in the organic liquid phase. However, using  
 MeCN as medium polar solvent with a low evaporation  
 temperature enabled us to isolate small particles on the  
 Au(111) surface, which should consist of only very few or even  
 single  $V_{18}$  polyoxoanions. Further experiments with less polar  
 solvents as well as the usage of other deposition techniques will  
 be part of future studies.

**XPS Data.** The electronic properties of the  $V_{18}$  poly-  
 oxoanions adsorbed on the Au(111) surface were investigated  
 by XPS as a function of solvent and concentration. For the  
 measurements two significant solution concentrations of  
 compound **1** were used: a lower concentration of  $1.6 \times 10^{-3}$   
 $\text{g}\cdot\text{L}^{-1}$ , which is in the range of our STM measurements showing  
 a submonolayer coverage, and a higher concentration of  $1.6 \times$   
 $10^{-2} \text{ g}\cdot\text{L}^{-1}$  resulting in a multilayer coverage. For comparison  
 also a solid-state XPS spectrum of the pure, unsolved  $V_{18}$   
 material was recorded. Figure 6 shows the corresponding  
 binding energy region for the O 1s, V 2p<sub>1/2</sub>, and V 2p<sub>3/2</sub> core  
 levels. After background subtraction three groups of peaks are  
 visible in the interesting binding energy region between  $510$   
 and  $540 \text{ eV}$ , which can be assigned to different oxygen and  
 vanadium species.

At highest energies, oxygen 1s peaks can be found at  $532.0$   
 and  $530.0 \text{ eV}$ , respectively. While the sharper peak at lower  
 energy can be clearly assigned to oxygen of  $V_{18}$ , the significantly  
 broader peak is located in the typical range of surface  
 adsorbates like hydroxide and carbonate species, which is  
 characteristic for ex situ prepared samples. At low  $V_{18}$   
 concentration this component dominates the core level spectra.  
 The two other groups of peaks belong to the 2p<sub>1/2</sub> and 2p<sub>3/2</sub>  
 features of different vanadium species.<sup>83</sup> While the peaks at  
 $524.2$  and  $516.7 \text{ eV}$  can be assigned to fully oxidized  $V^V$ , partly  
 reduced  $V^{IV}$  appears at  $523.0$  and  $515.5 \text{ eV}$ , respectively. For  
 the unmodified  $V_{18}$  structure we would expect a  $V^{IV}$  to  $V^V$   
 ratio of  $10:8$ . The measured relative vanadium contents based on the  
 integration of the 2p<sub>3/2</sub> peak intensities show for the bulk  
 measurement and both solvents with the higher  $V_{18}$   
 concentration a slight excess of  $V^{IV}$  within the expected range  
 indicating mainly unmodified  $V_{18}$  polyoxoanions (Table 4).



**Figure 6.** XPS spectra of the bulk  $V_{18}$  sample and a  $V_{18}$  covered Au(111) surface prepared by deposition from DMF or MeCN solution with different concentration of compound **1**. Peak fitting results in two O 1s peaks at 532.0 eV (gray curve) and 530.0 eV (green curve). Additionally, two different V species,  $V^V$  (blue curve) and  $V^{IV}$  (red curve), are detected with fitted  $2p_{1/2}$  peaks at 524.2 and 523.0 eV, and  $2p_{3/2}$  peaks at 516.7 and 515.5 eV, respectively.

**Table 4.** Data from XPS Spectra Shown in Figure 6

solvent (concentration)	vanadium $2p_{3/2}$	fitted peak position [eV]	relative vanadium content [%]
bulk	$V^{IV}$	515.4	7.0
	$V^V$	516.7	5.1
DMF ( $1.6 \times 10^{-2} \text{ g}\cdot\text{L}^{-1}$ )	$V^{IV}$	515.5	7.9
	$V^V$	516.7	5.9
MeCN ( $1.6 \times 10^{-2} \text{ g}\cdot\text{L}^{-1}$ )	$V^{IV}$	515.5	6.6
	$V^V$	516.7	6.0
MeCN ( $1.6 \times 10^{-3} \text{ g}\cdot\text{L}^{-1}$ )	$V^{IV}$	515.6	2.3
	$V^V$	516.7	4.8

635 The ratio changes drastically for the lower  $V_{18}$  concentration  
636 toward a higher amount of  $V^V$ . Although in this case the results  
637 should be interpreted with caution due to the overall low  
638 vanadium signal intensity, this observation indicates a partly  
639 oxidation of the mixed-valence  $V_{18}$  polyoxoanions on the  
640 surface.

641 The change in the vanadium ratios might be explained by the  
642 different binding situation of the  $V_{18}$  polyoxoanions on the gold  
643 surface. Using a concentration of  $10^{-3} \text{ g}\cdot\text{L}^{-1}$  leads to a  
644 submonolayer coverage as shown by the STM measurements.  
645 Thus, most  $V_{18}$  polyoxoanions have direct contact to the  
646 metallic substrate, which induces a change of the POV's redox  
647 state by a charge transfer process. Using a higher concentrated  
648 solution of  $10^{-2} \text{ g}\cdot\text{L}^{-1}$  for deposition results in a multilayer  
649 coverage, where the majority of the  $V_{18}$  polyoxoanions has no  
650 direct contact to the metal, so that their electronic structure  
651 remains unaffected.

## 652 ■ CONCLUSIONS

653 By combining physical and chemical methods with locally  
654 resolved surface science techniques and MD simulations, it was  
655 possible to link the behavior of a highly charged  $V_{18}$  POV in  
656 solution and after adsorption on a metallic surface. Hereby, we

demonstrated that the proneness of POVs to agglomerate in 657  
the liquid phase is strongly dependent on the polyoxoanion... 658  
polyoxoanion interface that is governed by the solution 659  
medium (water vs organic solvent vs a mixture of water and 660  
organic solvent). The role of the counteranion...solvent affinity 661  
in such agglomeration processes should not be neglected. 662  
Remarkably, the ion-pairing induced agglomeration of the  $V_{18}$  663  
POVs increases significantly upon molecular deposition of the 664  
bulk sample **1** from organic solution onto the Au(111) surface. 665  
The size and shape of agglomerates and the strength of their 666  
binding to the metallic substrate are controlled by the physical 667  
properties such as polarity and boiling temperature of the 668  
organic solvent (MeCN vs DMF). While the solvent polarity 669  
and the resulting charge stabilization is decisive for agglomeration 670  
in solution, this effect is subordinated for the adsorption 671  
behavior. In that case, surface mobility is the key factor for 672  
agglomeration, which can be enhanced by a lower evaporation 673  
speed of the solvent. A higher amount of available hydrogen 674  
ions as a function of the water content increases the 675  
agglomeration tendency of the vanadium-oxo clusters in 676  
solution at higher water concentrations. Small water impurities, 677  
caused, for example, by humidity, have no significant influence. 678

These observations provide guidelines for fine-tuning the 679  
POV-surrounding (counterions and solvent molecules) rela- 680  
tionships in solution as well as for the adjustment of the 681  
molecular adsorption on surfaces, which furthermore may open 682  
doors for potential photoluminescent effects<sup>84,85</sup> of POV 683  
particles on a solid support. This is worth further exploring 684  
because we have shown that the electronic structure of 685  
adsorbed POVs on the metallic surface undergoes redox 686  
modifications by the alteration of the molecule concentration- 687  
modulated surface coverage. Thus, it can be concluded that the 688  
interaction with the surface strongly influences the redox 689  
activity of the  $V_{18}$  polyoxoanions, which was detected by our 690  
CV measurements for the quasi-isolated vanadium-oxo clusters 691  
in solution. 692

The high complexity between the POV behavior in solution 693  
and on surfaces as shown in our study remains a key challenge 694  
for the utilization of this highly promising material class. The 695  
thermal controlled deposition on surfaces will be another 696  
logical step in order to fathom the possibilities for integrating 697  
POVs in real catalytic systems and technological devices. This is 698  
currently being explored in our laboratories. 699

## ■ ASSOCIATED CONTENT

### 📄 Supporting Information

The Supporting Information is available free of charge on the 702  
ACS Publications website at DOI: 10.1021/acs.jpcc.7b02138. 703

Experimental data of compound **1** including elemental 704  
analysis, powder X-ray diffractogram, UV/vis and IR 705  
spectra, cyclic voltammograms, and additional SAXS 706  
data; details of MD calculations (PDF) 707

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### Notes

The authors declare no competing financial interest. 714  
715

## 716 ■ ACKNOWLEDGMENTS

717 This work was supported by the Emmy Noether program of  
718 the Deutsche Forschungsgemeinschaft (DFG). K.Y.M. and  
719 M.M. are grateful to the Jülich-Aachen Research Alliance–  
720 Future Information Technology (JARA-FIT) for a Seed Fund  
721 grant. X.L. gratefully acknowledges the European Science  
722 Foundation (PoCheMon COST action) for funding of a short-  
723 term scientific mission at the RWTH Aachen University, the  
724 Government of Catalonia (project 2014SGR199), and the  
725 Spanish Ministerio de Ciencia e Innovación (project  
726 CTQ2014-52774-P). We also thank Prof. Dr. Claire Besson  
727 (Department of Chemistry, The George Washington Uni-  
728 versity) for helpful discussions and suggestions.

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